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# Dirhodium Triyttrium Disilicide, $\mathbf{Y}_{\mathbf{3}} \mathbf{R h}_{\mathbf{2}} \mathbf{S i}_{\mathbf{2}}$ 

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#### Abstract

M_{r}=528.7, \quad P b c m, \quad a=5.5661\) (5), $\quad b=$ 7.7171 (5), $c=13.043$ (1) $\AA, V=560.26 \AA^{3}, Z=4$, $D_{x}=6.26 \mathrm{~g} \mathrm{~cm}^{-3}, \quad$ Мо $K \alpha, \quad \lambda=0.71069 \AA, \quad \mu=$ $376 \cdot 1 \mathrm{~cm}^{-1}, F(000)=940$, room temperature, final $R=0.045$ for 678 unique reflections. The structure is made of Y and Si trigonal prisms centred by Si and Rh atoms and can be derived from a hexagonal closepacked model using the concept of unit-cell twinning. The stacking of these units is the same as in binary alloys with $\mathrm{Fe}_{3} \mathrm{C}$ - or $\mathrm{Mn}_{5} \mathrm{C}_{2}$-type structures.

Introduction. Structures of $R_{3} M$ binary alloys of $\mathrm{Fe}_{3} \mathrm{C}$, $\mathrm{Mn}_{5} \mathrm{C}_{2}$ and CrB type have been studied (e.g. Moreau, Le Roy \& Paccard, 1982). These structures can be derived from a hexagonal close-packed model through twinning of unit cells (Andersson \& Hyde, 1974). Here, the study is extended to the three-component alloy $\mathrm{Rh}_{2} \mathrm{Y}_{3} \mathrm{Si}_{2}$.


Experimental. Alloys made from commercially available elements of high purity ( $\mathrm{Y}: 99.9 \%$, Rh: $99.99 \%$, Si $: 99.999 \%$ ). Samples prepared by conventional arcmelting techniques. Small crystals of $\mathrm{Y}_{3} \mathrm{Rh}_{2} \mathrm{Si}_{2}$ suitable for X-ray analysis isolated by mechanical fragmentation from crushed melt. Weissenberg photographs showed crystals to have space group $P b c m$ or $P b c 2_{1}$. Single crystal $40 \times 60 \times 80 \mu \mathrm{~m}$, intensities measured with graphite-monochromated Mo $K \alpha$ radiation, CAD-4 Enraf-Nonius diffractometer, $\theta / 2 \theta$ scan mode; accurate cell dimensions by least-squares
analysis of $25 \theta$ values; three standard reflections monitored at intervals of 3600 s, intensity variation during data collection $<1 \%$. Intensities of 1275 independent reflections, to limit of $\theta=35^{\circ}$, measured and corrected for background, Lorentz and polarization factors with START (Frenz, 1983). $0 \leq h \leq 8$, $0 \leq k \leq 12, \quad 0 \leq l \leq 20$. Structure solved by direct methods with MULTAN (Main, Fiske, Hull, Lessinger, Germain, Declercq \& Woolfson, 1980). Positions of Y, Rh and Si atoms revealed in corresponding $E$ map for space group Pbcm . Full-matrix least-squares refinement of 36 positional and anisotropic thermal parameters converged after a few cycles using $F$ values of 678 reflections* with $I>3 \sigma(I) . R=R_{w}=0.045$, Dunitz \& Seiler (1973) weighting scheme where $w=\exp (36 \sin \theta / \lambda)$ for each reflection. Zero $\Delta / \sigma$ in final least-squares cycle; peak height in final difference map $<15.0 \mid \mathrm{e} \AA^{-3}$. All calculations performed on PDP 11/23 computer using SDP (Frenz, 1983). Atomic scattering factors from International Tables for X-ray Crystallography (1974).

Atomic positions and equivalent isotropic temperature factors are listed in Table 1, interatomic distances are in Table 2.

[^0]Table 1. Atomic positions for $\mathrm{Y}_{3} \mathrm{Rh}_{2} \mathrm{Si}_{2}$ with e.s.d.'s in parentheses

| $B_{\text {eq }}$ is defined as $\frac{4}{3} \sum_{i} \sum_{j} B_{i j} \mathbf{a}_{i} \cdot \mathbf{a}_{j}$. |  |  |  |  |
| :---: | :---: | :---: | :---: | :---: |
|  | $x$ | $y$ | $z$ | $B\left(\AA^{2}\right)$ |
| Y(1) | 0.1377 (2) | 0.6057 (1) | $0 \cdot 10313$ (6) | 0.785 (8) |
| Y(2) | 0.3713 (3) | 0.2472 (2) | $\frac{1}{4}$ | 0.76 (1) |
| Rh | 0.6361 (1) | 0.46771 (9) | $0 \cdot 09020$ (5) | 0.756 (5) |
| $\mathrm{Si}(1)$ | 0.8584 (8) | 0.3976 (6) | $\frac{1}{4}$ | 0.77 (4) |
| Si(2) | 0.3836 (7) | 1 | 0 | 0.75 (4) |

Table 2. Interatomic distances in $\mathrm{Y}_{3} \mathrm{Rh}_{2} \mathrm{Si}_{2}$ up to $4.0 \AA$

|  |  |  |  |
| :---: | :---: | :---: | :---: |
|  | All e.s.d.'s are less than $0.001 \AA$. |  |  |
| $\mathrm{Y}(1)-\mathrm{Rh}$ | 2.875 | $\mathrm{Y}(2)-\mathrm{Si}(1)$ | 2.949 |
| $-\mathrm{Si}(1)$ | 2.944 | $-\mathrm{Si}(1)$ | 2.985 |
| $-\mathrm{Si}(1)$ | 2.957 | -2 Rh | 3.000 |
| -Rh | 2.875 | -2 Rh | 3.068 |
| -Rh | 2.993 | $-\mathrm{Si}(1)$ | 3.082 |
| -Rh | 3.069 | $-2 \mathrm{Si}(2)$ | 3.262 |
| $-\mathrm{Si}(2)$ | 3.186 | $-2 \mathrm{Y}(1)$ | 3.512 |
| $-2 \mathrm{Si}(2)$ | 3.386 | $-2 \mathrm{Y}(1)$ | 3.590 |
| $-\mathrm{Y}(1)$ | 3.492 | $-2 \mathrm{Y}(1)$ | 3.607 |
| $-\mathrm{Y}(1)$ | 3.500 | $\mathrm{Rh}-\mathrm{Si}(2)$ |  |
| $-\mathrm{Y}(2)$ | 3.512 | $-\mathrm{Si}(1)$ | 2.478 |
| $-\mathrm{Y}(2)$ | 3.590 | $-\mathrm{Si}(2)$ | 2.483 |
| $-\mathrm{Y}(2)$ | 3.607 | -Rh | 2.486 |
| $-\mathrm{Y}(1)$ | 3.831 | $-\mathrm{Y}(1)$ | 2.842 |
|  |  | $-\mathrm{Y}(1)$ | 2.875 |
| $\mathrm{Si}(1)-2 \mathrm{Rh}$ | 2.483 | $-\mathrm{Y}(1)$ | 2.976 |
| $-2 \mathrm{Y}(1)$ | 2.944 | $-\mathrm{Y}(2)$ | 2.993 |
| $-\mathrm{Y}(2)$ | 2.949 | $-\mathrm{Y}(2)$ | 3.000 |
| $-2 \mathrm{Y}(1)$ | 2.957 | $-\mathrm{Y}(1)$ | 3.068 |
| $-\mathrm{Y}(2)$ | 2.985 |  | 3.069 |
| $-\mathrm{Y}(2)$ | 3.082 | $\mathrm{Si}(2)-2 \mathrm{Rh}$ | 2.478 |
|  |  | $-2 R h$ | 2.486 |
|  |  | $-2 \mathrm{Y}(1)$ | 3.186 |
|  |  | $-2 \mathrm{Y}(2)$ | 3.262 |
|  |  | $-2 \mathrm{Y}(1)$ | 3.349 |
|  |  |  |  |
|  |  |  |  |



Fig. 1. Projection along the $a$ axis for $\mathrm{Y}_{3} \mathrm{Rh}_{2} \mathrm{Si}_{2}$. Numbers correspond to $x$ parameters multiplied by 100 . Drawing of trigonal prisms becomes significant if 100 is added to inscribed values of $100 x$ for upper atoms and if 100 is subtracted for lower atoms. Large circles are Y atoms, small circles Rh atoms and filled circles Si atoms.

Discussion. The structure of $\mathrm{Y}_{3} \mathrm{Rh}_{2} \mathrm{Si}_{2}$ is shown in Fig. 1 in projection along the $a$ axis. All Y atoms participate in the formation of trigonal prisms centred by Si atoms. On each side of the central Y prism, there is a prism made of one common face of four Y atoms and one edge of two Si atoms. These two adjacent prisms are centred by one Rh atom. The $\mathrm{Y}_{3} \mathrm{Rh}_{2} \mathrm{Si}_{2}$ structure involves units of three adjacent trigonal prisms centred by Si and Rh atoms. The linkage of these units along the $b$ direction and the stacking along the $a$ direction are identical to those found in binary alloys crystallizing with the $\mathrm{Fe}_{3} \mathrm{C}$ or $\mathrm{Mn}_{5} \mathrm{C}_{2}$ type structures (Parthe \& Moreau, 1977). Also indicated on Fig. 1 are zig-zag chains of atoms along the $c$ axis. It has been shown that these chains can be geometrically derived from a hexagonal close-packed structure (Andersson \& Hyde, 1974). Twinning along particular planes leads to the formation of trigonal prismatic holes on the twin planes. Structures of $\mathrm{YPd}_{2} \mathrm{Si}$ (Moreau, Le Roy \& Paccard, 1982) of $\mathrm{Fe}_{3} \mathrm{C}$ type and $R_{5} \mathrm{Ir}_{2}(R=$ rare earth) of $\mathrm{Mn}_{5} \mathrm{C}_{2}$ (Le Roy, Paccard \& Moreau, 1980) and CrB type (Parthé, 1976) have been interpreted using this concept of unit-cell twinning. In $\mathrm{Y}_{3} \mathrm{Rh}_{2} \mathrm{Si}_{2}$ the hexagonal close-packed structure is made of Y and Si atoms and the prismatic holes are occupied by Rh and Si atoms. In $\mathrm{YPd}_{2} \mathrm{Si}$ the hexagonal close-packed structure was made of Y and Pd atoms and the prismatic holes were occupied by Si atoms only. The radii of metallic atoms given by Teatum, Gschneider \& Waber (1960) are 1.319 for $\mathrm{Si}, 1.345$ for Rh and $1.376 \AA$ for Pd , so that it is not surprising to find both Rh and Si atoms either in prismatic holes or in the chains.

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[^0]:    * Lists of structure factors and anisotropic thermal parameters have been deposited with the British Library Lending Division as Supplementary Publication No. SUP 39420 ( 9 pp.). Copies may be obtained through The Executive Secretary, International Union of Crystallography, 5 Abbey Square, Chester CHI 2HU, England.

